

LIFE PREDICTION AND CONSTITUTIVE MODELS FOR
ENGINE HOT SECTION ANISOTROPIC MATERIALSGUSTAV A. SWANSON
Pratt & Whitney
United Technologies Corporation

INTRODUCTION

The development of directionally solidified and single crystal alloys is perhaps the most important recent advancement in hot section materials technology. By reducing or eliminating grain boundaries in superalloys, the high temperature strengths have been substantially improved. Metallurgists have developed the alloy chemistries and casting processes so that they are now in practical use. However, the life limits of gas turbine parts, under complex loading conditions, are still not well known or understood. The objective of this program is to develop that knowledge to enable the designer to improve anisotropic gas turbine parts to their full potential.

Program Overview

The base program, which is followed by two options (fig. 1), will concentrate on coated turbine blade airfoil conditions. The coating, which is added to turbine airfoils to improve their oxidation and corrosion life, plays a major role in fatigue initiation. For this reason coated specimens are used extensively in this program. The materials have been selected, specimen fabrication is underway, the literature search is completed and Task III testing has begun. Table I shows the task breakdown of the base program.

Material Selection

The two single crystal alloys selected are PWA 1480 and Alloy 185. Table II lists the composition of these alloys. PWA 1480 was selected because it is the single crystal alloy most widely used today in gas turbine engines. Furthermore, it is representative of other practical single crystal alloys. Alloy 185 was selected because of differences between it and PWA 1480. These differences include a high volume fraction of γ' due to the higher aluminum content, a large γ/γ' misfit due to the high molybdenum content, and a higher level of creep anisotropy at higher temperatures. Contrast between the two alloys will provide a good test of the generality of any life prediction and constitutive models developed in this program.

The coatings selected are an overlay coating, PWA 286, and an aluminide diffusion coating, PWA 273. Coating chemistries are listed in Table III. These widely used coatings represent two basic classes.

Test Specimens

The constitutive specimens are solid and cylindrical; the fatigue specimens are hollow and cylindrical. The latter geometry is particularly applicable to thermo-mechanical fatigue (TMF) specimens. The thin wall facilitates thermal transient responses at reasonably fast rates. In addition, compressive stresses can be achieved

with minimum risk of buckling. The crystallographic orientation of the specimens for both the constitutive and life prediction program will include $\langle 100 \rangle$, $\langle 110 \rangle$, $\langle 111 \rangle$, and $\langle 123 \rangle$. Under tensile loading, the first two orientations produce slip along the orthogonal planes; the third along cuboidal planes, and the fourth a mixture of the two.

Test specimens for the coatings' tensile, constitutive, and life properties required special preparation. The plasma-sprayed specimens were fabricated by two methods; some were machined from HIPed bulk powder. The others were sprayed with a 1.5 mm (0.060 in.) layer of coating on a metal substrate which was subsequently removed by machining. On turbine blades the plasma-sprayed coating is about 0.4 mm (0.015 in.) thick and finished with shot peen. Metallographically the two specimens bracket the actual porosity of airfoil coating (see fig. 2).

Diffusion coating properties are impossible to measure directly. The strategy in this program will be to utilize two thicknesses of substrate. Specimens of both thicknesses will be coated and then tested for tensile, creep and fatigue properties. The results will then be plotted versus substrate thickness and extrapolated to zero thickness to obtain the values for the coating alone.

Models to be Evaluated

An extensive literature search has been completed for both the constitutive and life prediction models. The bulk of the past work has been done on isotropic materials. This research will be adapted to anisotropic materials whenever possible. For example, the life prediction models may use maximum resolved shear strain ranges on the active slip planes in place of principle strain ranges. The constitutive models to be considered will include macroscopic continuum theories of Hill (ref. 1), and Lee and Zaverl (ref. 2 and 3), or a unified visco-plastic formulation such as that of Walker and Cassenti (ref. 4 and 5). The unified theory is currently being extended by Walker to recognize specific slip systems of nickel-based single crystal alloys. Stouffer, in a parallel program (ref. 6), is also developing a constitutive model for single crystal alloys. All of these models will be evaluated utilizing the test data generated in this program.

Life prediction models under consideration include a number of isotropic models: linear time-cycle fraction (ref. 7), ductility exhaustion (ref. 8 and 9), frequency modified life (ref. 10, 11 and 12), frequency separation (ref. 13 and 14), Ostergren's method (ref. 15), strain range partitioning (ref. 16), damage mechanics, (ref. 17 and 18), continuous damage (ref. 19 and 20), and cumulative damage approach (ref. 21). All of these models will be evaluated utilizing resolved shear stress and/or strain on active slip planes. A model to study the interaction of the coating and substrate is being developed.

Test Results

Because the test program is in its early stages evaluation of the results would be premature. One interesting preliminary result, however, is a comparison of the fracture surfaces of tensile specimens of PWA 1480 pulled in the $\langle 001 \rangle$ direction shown in figure 3. Note that the faceting at 760°C (1400°F) is pronounced and that

the number of active slip planes is small. However, as the test temperature is increased to 1093°C (2000°F), the number of faceting planes becomes more numerous and the fracture surface appears more normal to the tensile load.

Plans

By October 1985, we plan to complete the Task III tests and to have a preliminary evaluation of the constitutive and life prediction models.

REFERENCES

1. Hill, R.: A Theory of the Yielding and Plastic Flow of Anisotropic Metals. Proc. Royal Society of London, Ser. A., Vol. 193, pp. 281-297, 1948.
2. Lee, D.; Zaverl, F. Jr.; Shih, C. F.; and German, M. D.: Plasticity Theories and Structural Analysis of Anisotropic Metals. Report No. 77CRD285, General Electric Corporate Research and Development Center. Schenectady, New York, 1977.
3. Lee, D.: Anisotropic Yielding Behavior of a Fiber-Reinforced, Directionally Solidified Eutectic Alloy. Metallurgical Transactions A, Vol. 9A, pp. 1477-1481, 1978.
4. Walker, K. P.: Research and Development Program for Nonlinear Structural Modeling With Advanced Time-Temperature Dependent Constitutive Relationships. NASA CR-165533, November 1981.
5. Cassenti, B. N.: Follow on program to Reference 4. NASA Lewis Contract, NAS3-23273.
6. Stouffer, D.: Evaluation of Constitutive Modeling Method for Single Crystal Superalloys. Third Annual Workshop, Turbine Hot Section Technology (HOST) Project, NASA Lewis Research Center, Cleveland, Ohio, Oct. 23-24, 1984.
7. Class 1 Components in Elevated Temperature Service, Class III. ASME Boilers and Pressure Vessel Code, Case Interpretations. Code Case 1592, American Society of Mechanical Engineers, N. Y.
8. Sach, G.; Gerberich, W. W.; Weiss, V.; and Lavorre, J. V.: Proc. ASTM, V. 60, 1961, pp. 512-529.
9. Polhemus, J. F.; Spaeth, C. E.; Vogel, W. H.: Ductility Exhaustion Model Prediction of Thermal Fatigue and Creep Interaction. Fatigue at Elevated Temperature, ASTM STP520, American Society for Testing and Materials, 1973, pp. 625-636.
10. Coffin, L. F. Jr.: Fatigue at High Temperature. Fatigue at Elevated Temperatures, ASTM STP 520, American Society for Testing and Materials, 1973, pp. 5-34.

11. Coffin, L. F., Jr.: Fatigue at High Temperature Prediction and Interpretation. James Clayton Lecture, Institute of Mechanical Engineers, Vol. 188, 1974, pp. 109-127.
12. Coffin, L. F., Jr.: The Effect of Frequency on Cyclic Strain and Fatigue Behavior of Cast Rene at 1600°F. Metallurgical Transactions, Vol. 5, May 1974, pp. 1053-1060.
13. Coffin, L. F.: The Concept of Frequency Separation in Life Prediction for Time-Dependent Fatigue. Symposium on Creep Fatigue Interaction, 1976, pp. 349-363.
14. McKnight, R. L.; Lafen, J. H.; and Spmer, G. T.: Turbine Blade Tip Durability Analysis. National Aeronautics and Space Administration Report, NASA CR-165268.
15. Ostergren, W. J.: A Damage Function and Associated Failure Equation for Predicting Hold Time and Frequency Effects in Elevated Temperature, Low Cycle Fatigue. Journal of Testing and Evaluation, JTEVA, Vol. 4., No. 5, September 1976, pp. 327-339.
16. Manson, S. S.; Halford, G. R.; and Nachtigall, A. C.: Separation of the Strain Components for Use in Strain Range Partitioning. Symposium on Advances in Design for Elevated Temperature Environment, American Society for Mechanical Engineers, 1975, pp. 17-28.
17. Chaboche, J. L.: Thermodynamic and Phenomenological Description of Cyclic Viscoplasticity with Damage. European Space Agency Technical Translation, May 1979.
18. Lemaitre, J.; and Chaboche, J. L.: A Nonlinear Model of Creep Fatigue Damage and Accumulation and Interaction. Symposium at IUTAM, Sur La Mecanique Des Milieux et Des Corp Viscoelastiques, Gothenburg, Sweden, September 2-6 1974.
19. Majumdar, S.; and Miaya, P. S.: Wave Shaped Effects in Elevated Temperature Low-Cycle Fatigue on Type 304 Stainless Steel. T. Y. Chang and E. Krempl (EDS.), Inelastic Behavior of Pressure Vessel and Piping Components, PVP-PD-028, ASME, New York, pp. 43-54.
20. Majumdar, S.: Designing Against Low Cycle Fatigue at Elevated Temperature. Nuclear Engineering and Design, Vol. 63, 1981, pp. 121-135.
21. Moreno V.: NASA Contract NAS 3-23288

TABLE I
BASE PROGRAM TASKS

- I Material/Coating Selection and Acquisition
- II Selection of Candidate Life Prediction and Constitutive Models
- III Level 1 Experiments
- IV Correlation of Models With Level 1 Single Crystal Experiments
- V Level 2 Single Crystal Experiments
- VI Final Selection of Life Prediction and Constitutive Models
- VII Subcomponent Verification For Primary Single Crystal Material
- VIII Alternate Single Crystal Material Characterization For Airfoil Applications
- IX Model Verification On Alternate Single Crystal Material
- X Delivery of Computer Code to NASA

TABLE II
SINGLE CRYSTAL ALLOY COMPOSITION
(Weight Percent)

<u>Alloy</u>	Ni	Cr	Co	Ti	Al	Ta	W	Mo	Nb	C	B	Zr	Hf	Y
PWA 1480	Bal*	10.0	5.0	1.5	5.0	12.0	4.0	--	--	--	--	--	--	--
Alloy 185	Bal	--	--	--	6.8	--	6.0	14.0	--	0.04	--	--	--	--

* Balance

TABLE III
COATING MATERIALS

<u>Coating</u>	<u>Composition</u>	<u>Deposition Process</u>
Overlay PWA 286	NiCoCrAlY + Si+Hf	Vacuum Plasma Spray
Diffusion PWA 273	Aluminate/ Outward Diffusion	Gas Phase

PROGRAM OUTLINE

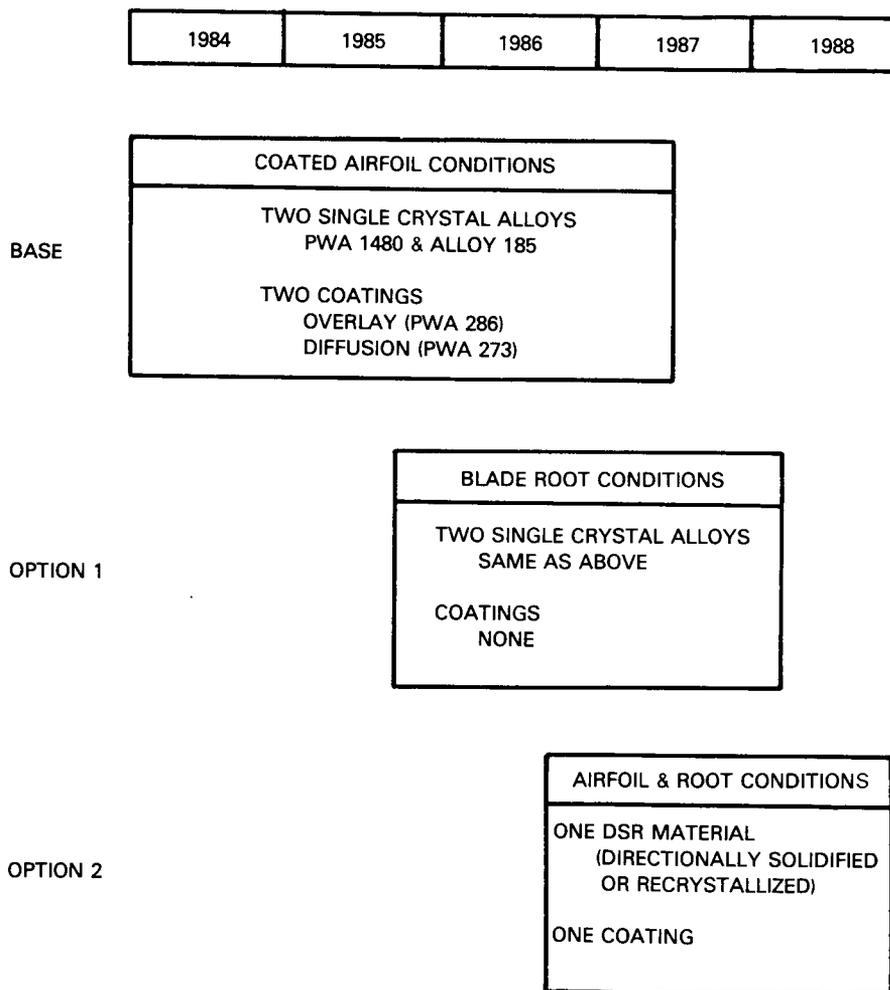
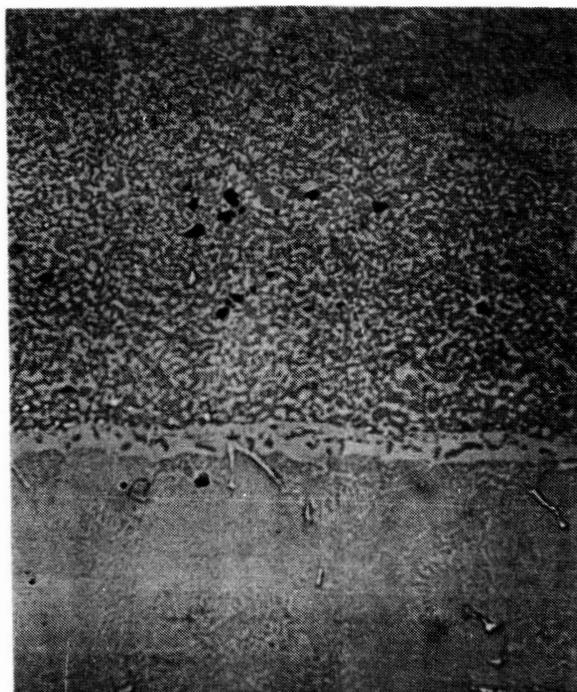


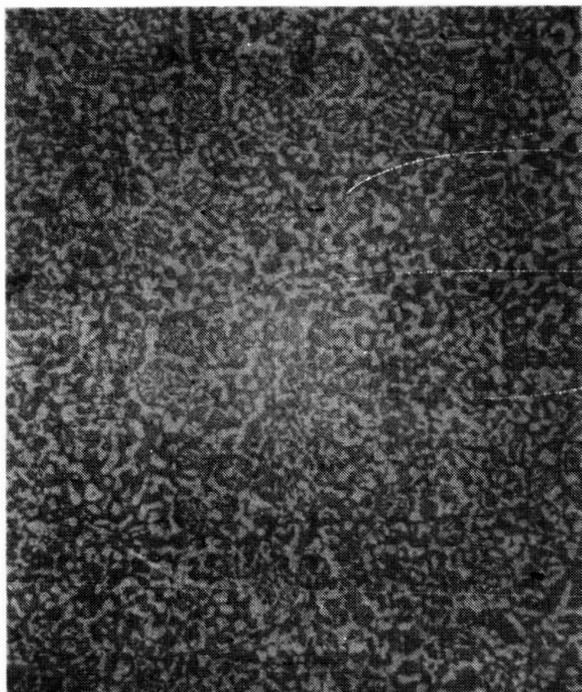
Figure 1

PWA 286 OVERLAY COATING STRUCTURE: STAND-ALONE vs PRODUCTION



ORIGINAL PAGE IS
OF POOR QUALITY

TYPICAL PRODUCTION COATING 500X
HEAT TREATED & PEENED



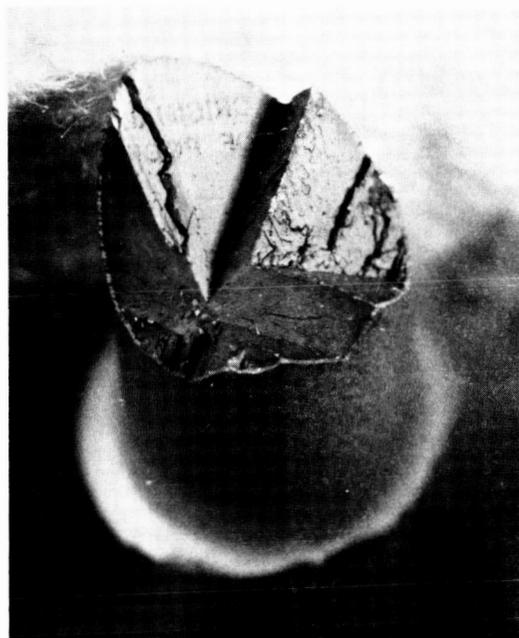
HIPed BULK POWDER 500X



THICK PLASMA SPRAY 500X
HEAT TREATED & PEENED

Figure 2

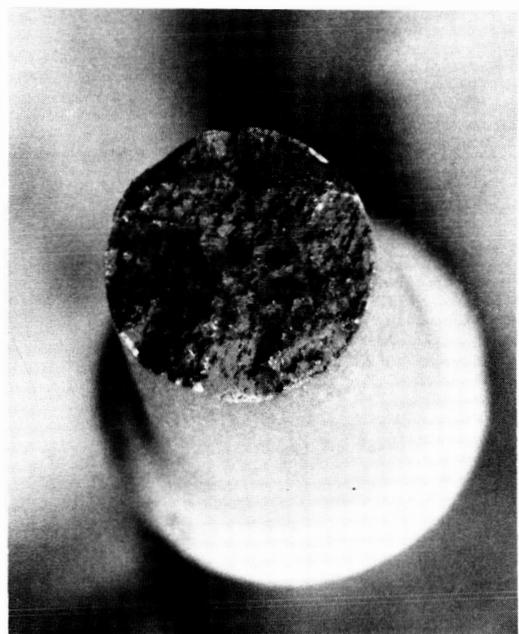
FRACTURE SURFACE OF PWA 1480 TENSILE SPECIMENS



760°C (1400°F)



871°C (1600°F)



982°C (1800°F)



1093°C (2000°F)

Figure 3